

Scientific paper

# The Effect of Temperature and Fe<sup>3+</sup> Concentration on the Formation of $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> Nanoparticles Embedded in Silica Matrix

Oana Ștefănescu,<sup>1,\*</sup> Corneliu Davidescu<sup>1</sup> and Paul Barvinschi<sup>2</sup><sup>1</sup> Faculty of Industrial Chemistry and Environmental Engineering, University „Politehnica”, P-ta Victoriei no.2, 30006, Timisoara, Romania<sup>2</sup> Faculty of Physics, West University, Bv. V. Parvan no. 4, 300223 Timisoara, Romania

\* Corresponding author: E-mail: oana.stefanescu@chim.upt.ro

Received: 13-11-2009

## Abstract

The paper presents a study on the formation and thermal stability of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles within the silica matrix depending on the Fe(III) carboxylate-type precursors, their annealing temperature and the concentration of Fe<sup>3+</sup>. Obtaining of the precursors, within the pores of the gel, bases on the redox reaction between Fe(NO<sub>3</sub>)<sub>3</sub> and diols: ethylene glycol (EG) and 1,4 buthane diol (1,4 BG), respectively. Thus, we have prepared gels with different Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub> ratios (20, 30, 50, 70 wt% Fe<sub>2</sub>O<sub>3</sub>) which were annealed in the temperature range 573-1273 K.

The formation and stability of the single  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> phase within the pores is strongly influenced by the reducing atmosphere generated upon thermal decomposition of the Fe(III) carboxylates. The XRD analysis evidenced a much stronger crystallization of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> in case of the nanocomposites synthesized with 1,4 BG than with EG.

The magnetic measurements confirm the crystallinity of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> within the silica matrix.

**Keywords:** Carboxylates,  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>, silica matrix, thermal analysis, XRD

## 1. Introduction

In the last years, the iron oxide-silica nanocomposites have been the subject of intense research due to their potential applications in magnetic-tape media,<sup>1</sup> magneto-optical devices,<sup>2</sup> magnetic refrigerators,<sup>3</sup> bioprocessing,<sup>4</sup> catalysis and ferrofluids.<sup>5,6</sup> The silica aerogels display most of the conditions required by the host matrix: chemical inertness, large surface areas, high porosity, pores in the nanometer range and high transparency.<sup>7</sup> The use of an inorganic host matrix for nanocrystalline particles could be an effective way for the limitation of particles size distribution and control on the homogenous dispersion of ultrafine metal oxides particles. Magnetic iron oxide nanoparticles exhibit enhanced surface effects, superparamagnetic behavior and changes in saturation magnetisation and coercive fields.<sup>8</sup>

Maghemite ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>) nanoparticles present unique magnetic, catalytic and optical properties compared to the

bulk material.<sup>9</sup> Embedding of magnetic particles within the silica matrix prevents their aggregation at temperatures up to 1273K and facilitates the stabilization of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>.<sup>10</sup> The sol-gel method is a promising way for obtaining of iron oxide nanoparticles within an ordered silica matrix. Silica and iron oxide precursors are mixed in solution and condensed by the sol-gel process in order to obtain a mixed metallic oxides network.

In previous papers, we have elaborated the synthesis method for some carboxylate-type complex combinations based on the redox reaction between transitional metal nitrates and diols.<sup>11,12</sup> Regarding the Fe<sub>2</sub>O<sub>3</sub> system obtained from the Fe(III) carboxylate type complexes, we have pursued the formation of the single  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> phase embedded in a SiO<sub>2</sub> matrix.<sup>13,14</sup>

The Fe(III) carboxylate-type complex combinations (glyoxylate [Fe<sub>2</sub>(C<sub>2</sub>H<sub>2</sub>O<sub>4</sub>)<sub>2</sub>(OH)<sub>2</sub>(H<sub>2</sub>O)<sub>2</sub>] and succinate [Fe<sub>2</sub>(C<sub>4</sub>H<sub>4</sub>O<sub>4</sub>)(OH)<sub>4</sub>(H<sub>2</sub>O)<sub>2</sub>] · 1.5 H<sub>2</sub>O), embedded in the pores of silica gels, generate a reducing atmosphere (CO/C) by thermal decomposition. The reducing atmo-

here achieves the in-situ redox equilibria  $\text{Fe}^{3+} \rightarrow \text{Fe}^{2+}$  and  $2 \text{Fe}^{2+} + 3/2 \text{O}_2 \rightarrow \text{Fe}_2\text{O}_3$ , respectively, responsible for the formation of the crystalline  $\gamma\text{-Fe}_2\text{O}_3$  phase. The composition of the reducing atmosphere depends on the nature of respective diols and influences the degree of crystallinity of the oxide  $\gamma\text{-Fe}_2\text{O}_3$  phase.

In this paper we present a study on the formation of the  $\gamma\text{-Fe}_2\text{O}_3$  nanocrystallites as the single phase within the  $\text{SiO}_2$  matrix using an original synthesis method (sol-gel), which allows the stabilisation of  $\gamma\text{-Fe}_2\text{O}_3$  at high temperatures. The formation of  $\gamma\text{-Fe}_2\text{O}_3$  depends on the Fe(III) carboxylate-type combination embedded in the pores of the silica gel (glyoxylate, succinate), the thermal treatment temperature (573–1273 K) and the mass ratio  $\text{Fe}_2\text{O}_3/\text{SiO}_2$  (20, 30, 50, 70 wt%).

## 2. Experimental

The gel samples were synthesized by the modified sol-gel method,<sup>15</sup> from a mixture of  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ , ethylene glycol (EG) and 1,4 butane diol (1,4 BG), respectively, tetra-ethyl orthosilicate (TEOS), ethanol as solvent and concentrated nitric acid ( $\text{HNO}_3$ ). All reagents were of high purity (>98%), product of Merck.

The ethanolic TEOS solution was added drop wise, under magnetic stirring, to the mixture  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ -diol (EG or 1,4 BG), for the ratios  $\text{Fe}_2\text{O}_3/\text{SiO}_2$ : 20, 30, 50, 70 wt%. **The molar ratio  $\text{NO}_3^-$ : diol used in the synthesis was:  $\text{NO}_3^-$ : EG = 1:0.75 and  $\text{NO}_3^-$ : 1,4 BG = 1: 0.56, representing 50% diol excess, related to the stoichiometry.** The samples were denoted as follows:  $E_x$  for gels synthesized with EG and  $B_x$  for gels synthesized with 1,4 BG, where  $x = 20, 30, 50, 70$  wt%. The obtained gels were dried at 313 K, 1h, and subsequently heated at 403K, 3 h, when the redox reaction between  $\text{NO}_3^-$  and diol took place with the formation of the Fe(III) carboxylate-type complex combination within the pores of the silica gel. The obtained powders were characterized by thermal analysis and FT-IR spectrometry.

The gels ( $E_x$  and  $B_x$ ) obtained at 403 K, were annealed in the temperature range 573–1273 K, in air, for 3 h, in a Nabertherm furnace, in order to obtain the oxides. The crystalline phases obtained in the composites were identified by X-ray diffraction on a D8 Advanced- Bruker AXS diffractometer, using Mo-  $K\alpha$  radiation ( $\lambda_{\text{Mo}} = 0.7093 \text{ \AA}$ ). **The transmission electron microscopy was performed with a JEOL JEM 1010 microscope.**

For monitoring the evolution of the redox reaction between Fe(III) nitrate and diols, the gels were characterized by thermal analysis using a 1500 D MOM Budapesta derivatograph. The thermal behaviour of the complex combinations embedded in the gels was followed by a Diamond Perkin Elmer thermo-balance in air, up to 773 K, with a heating rate of 10 K/min, with the sample mass of ~30 mg.

The samples were characterized by FT-IR spectrometry on a Shimadzu Prestige-21 a FT-IR spectrometer, in KBr pellets, in the range 400–4000  $\text{cm}^{-1}$ . Magnetic measurements were performed by a laboratory installation with the data acquisition system.

## 3. Results and Discussion

The formation of the Fe(III) carboxylate-type complex combinations within the pores of the gel was analyzed by thermal analysis and FT-IR spectrometry. Figure 1 presents the TG and DTA curves of the gel  $B_{30}$  heated at 313 K. The DTA curve presents an exothermic effect at 343 K attributed to the redox reaction between  $\text{Fe}(\text{NO}_3)_3$  and 1,4 BG with the formation of the Fe(III) complex combination (succinate) within the pores. The second

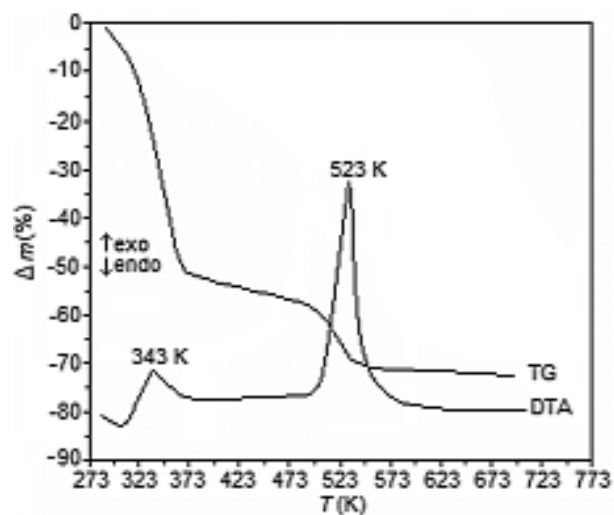


Figure 1. TG and DTA curves of the gel  $B_{30}$  heated at 313 K

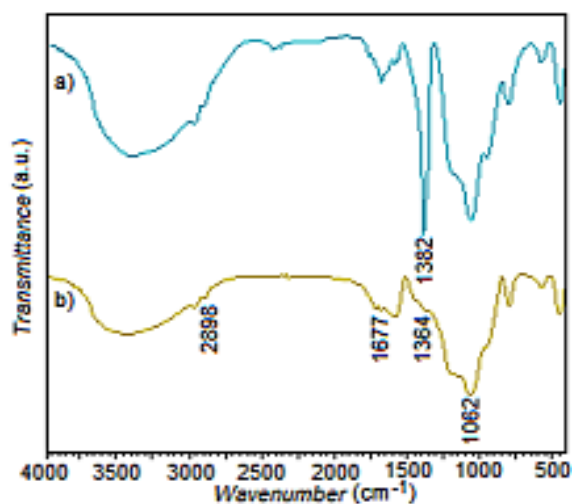


Figure 2. FT-IR spectra of the gel  $B_{30}$  heated at 313 K (a) and 403 K (b)

exotherm at 523 K corresponds to the oxidative decomposition of the formed complex combination. The mass losses on the TG, in the first stage, correspond to the elimination of volatile products:  $\text{H}_2\text{O}$ ,  $\text{NO}_x$ , and in the second stage, to the elimination of the oxidation products  $\text{CO}$ ,  $\text{CO}_2$  and condensation products of the matrix.

As a result of thermal analysis data, we have established 403 K as the optimal synthesis temperature for the complex combinations within the pores.

Figure 2 presents the FT-IR spectra of the gel  $\text{B}_{30}$  at 313 K (spectrum a) which evidences a clear band at  $1382\text{ cm}^{-1}$  attributed to  $\text{NO}_3^-$ , free in the pores of the gel. The

spectrum (b) corresponding to the gel thermally treated at 403 K shows the disappearance of the band at  $1382\text{ cm}^{-1}$  and the appearance of the bands characteristic for the complex combination in the pores:  $\nu_{\text{as}}(\text{COO}^-)$  at  $1677\text{ cm}^{-1}$  and  $\nu_{\text{s}}(\text{COO}^-)$  at  $1364\text{ cm}^{-1}$ .<sup>16</sup> In the range  $2800\text{--}3000\text{ cm}^{-1}$  (spectra a and b) the bands characteristic for the groups  $-\text{CH}_2-$ ,  $-\text{CH}_3$  from the carboxylates as well as for the diols chemically bonded within the matrix are registered.<sup>17</sup> The intense band at  $1062\text{ cm}^{-1}$  is attributed to the asymmetric stretching vibration  $\nu_{\text{as}}(\text{Si-O-Si})$ .<sup>18</sup>

All synthesized samples ( $\text{E}_x$ ,  $\text{B}_x$ ) present similar thermal and FT-IR behavior.

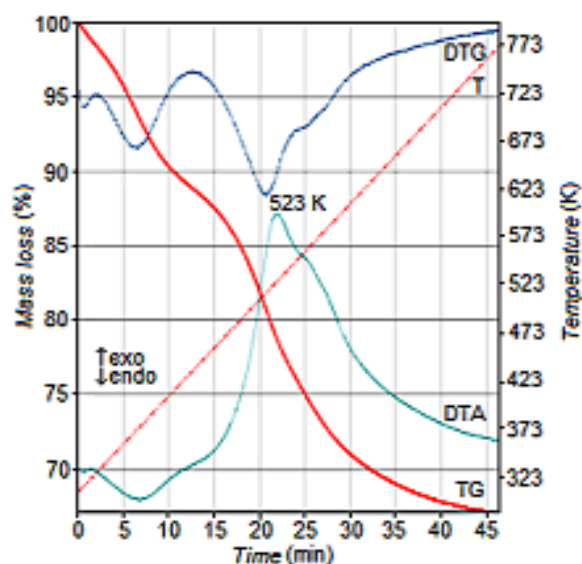


Figure 3. Thermal analysis curves of the gel  $\text{B}_{20}$  thermally treated at 403 K

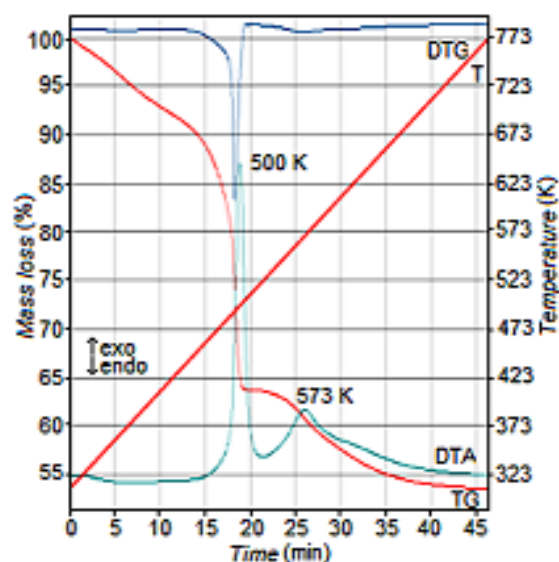


Figure 5. Thermal analysis curves of the gel  $\text{B}_{50}$  thermally treated at 403 K

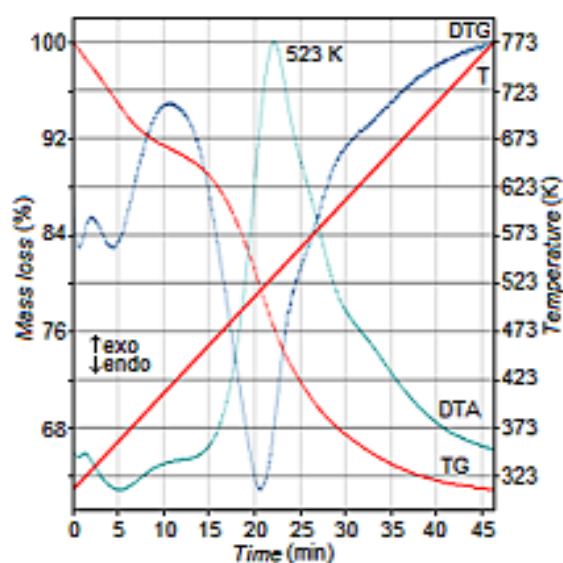


Figure 4. Thermal analysis curves of the gel  $\text{B}_{30}$  thermally treated at 403 K

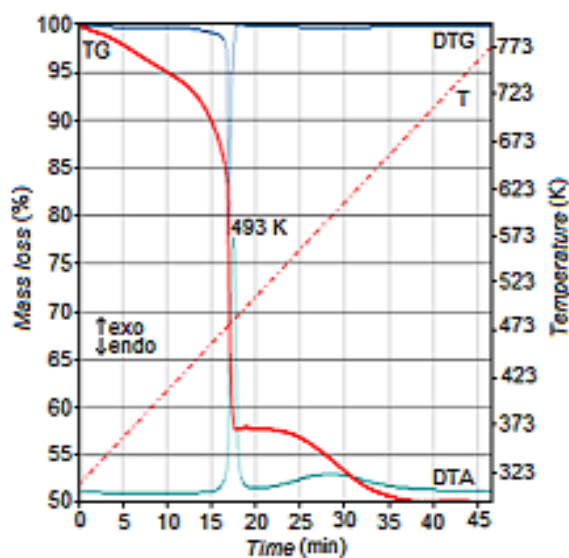


Figure 6. Thermal analysis curves of the gel  $\text{B}_{70}$  thermally treated at 403 K

Figures 3–6 present the thermal analysis curves for the gels  $B_x$  synthesized with 1,4 BG. The registered thermal processes justify the formation of the complex combination within the pores of the gels, in all cases. The evolution of the thermal analysis curves is influenced by the composition of the gels.

At low concentrations of 20 and 30 wt%  $Fe_2O_3/SiO_2$ , the Fe(III) complex combination is found in a lower quantity, uniformly distributed within the pores of the gels. The thermal decomposition process (Figures 3 and 4) with the elimination of CO, accompanied by a large exothermic effect at  $\sim 523$  K proceeds with a low rate. The

mass loss up to 773 K is attributed to the elimination of the poly-condensation products of the silica matrix.

At higher concentrations of 50 and 70 wt%  $Fe_2O_3/SiO_2$ , the decomposition of the complex combination (Figures 5 and 6) takes place at a higher rate in a narrower temperature range with a pronounced exothermic effect at  $\sim 500$  K. The mass loss in this range mostly corresponds to the oxidative decomposition of the complex combination. The weak exothermic effect at  $\sim 573$  K, with a mass loss, is attributed to the burning of the organic chain of the diol, bonded within the silica matrix during the poly-condensation process.<sup>17</sup>

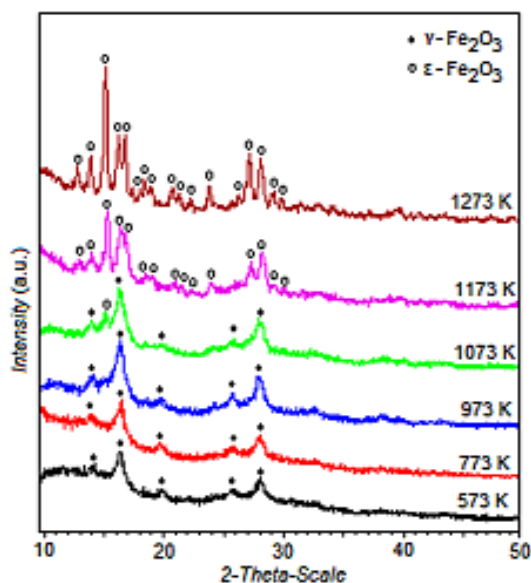


Figure 7. XRD patterns of the gel  $B_{50}$  annealed at different temperatures

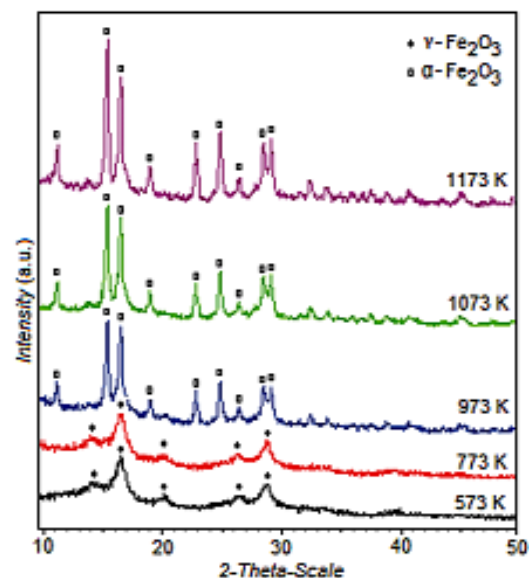


Figure 9. XRD patterns of the gel  $B_{70}$  annealed at different temperatures

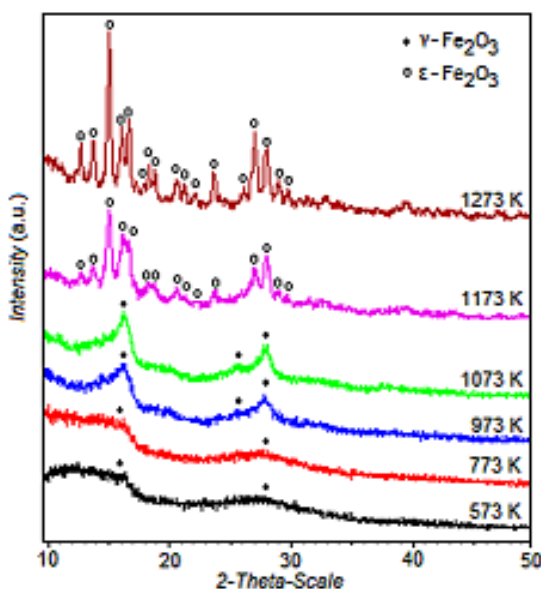


Figure 8. XRD patterns of the gel  $E_{50}$  annealed at different temperatures

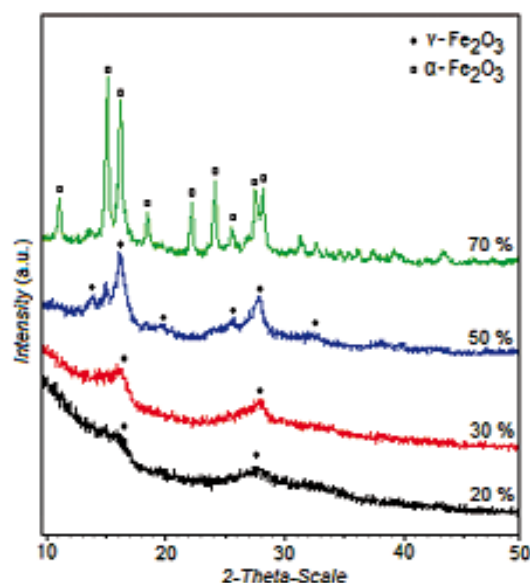


Figure 10. XRD patterns of gels  $B_x$  annealed at 1073 K

For all gels ( $B_x$ ,  $E_x$ ), the thermal decomposition process of the Fe(III) carboxylate type complex combination takes place within the pores of the gels with the generation of a reducing atmosphere (CO/C). The composition of the reducing atmosphere depends on the synthesized complex combination (on the nature of the diol). The reducing atmosphere (CO + 3C) generated upon the thermal decomposition of the Fe(III) succinate-type complex combination ( $[\text{Fe}_2(\text{C}_4\text{H}_4\text{O}_4)(\text{OH})_4(\text{H}_2\text{O})_2] \cdot 1.5 \text{ H}_2\text{O}$ ) strongly influences the formation and crystallization of the oxidic phase  $\gamma\text{-Fe}_2\text{O}_3$  compared to the less reducing atmosphere (4CO) generated by glyoxylate ( $[\text{Fe}_2(\text{C}_2\text{H}_2\text{O}_4)_2(\text{OH})_2(\text{H}_2\text{O})_2]$ ).

Figures 7–10 present the XRD patterns of gels  $B_x$  and  $E_x$  with formation of the single  $\gamma\text{-Fe}_2\text{O}_3$  phase within the pores of the silica matrix, depending on the annealing temperature, the precursor nature and the ratio  $\text{Fe}_2\text{O}_3/\text{SiO}_2$ .

In Figures 7 and 8 the XRD patterns of the samples  $B_{50}$  and  $E_{50}$  are presented, showing the evolution of the crystalline  $\text{Fe}_2\text{O}_3$  phases depending on the annealing temperature and the nature of the precursors. In XRD patterns in Figure 7 we can notice  $\gamma\text{-Fe}_2\text{O}_3$  as the single phase, crystallized within the pores of the matrix, at 573 K. This phase is maintained even at higher temperatures: 773, 973 and 1073 K. In case of the patterns from Figure 8 corresponding to sample  $E_{50}$ , thermally treated at the same conditions, the crystallinity of  $\gamma\text{-Fe}_2\text{O}_3$  at lower temperatures is weak. This can be a consequence of the less reducing atmosphere created at decomposition of the Fe(III) glyoxylate compared to Fe(III) succinate. The  $\gamma\text{-Fe}_2\text{O}_3$  is well crystallized at 973 and 1073 K.

In both cases, at 1173 K and 1273 K the patterns of the samples are essentially modified indicating the formation of a new phase identified as  $\epsilon\text{-Fe}_2\text{O}_3$  based on JCPDS 16–653 and the recent work of Brázda et al.<sup>19,20</sup>

In Figure 9 the XRD patterns of the gel  $B_{70}$  annealed at different temperatures are presented. The single  $\gamma\text{-Fe}_2\text{O}_3$  phase is well crystallized at 573 and 773 K. With increasing temperature, the single  $\alpha\text{-Fe}_2\text{O}_3$  phase crystallizes at 973 K and remains stable at 1073 and 1173 K. In case of this composition, due to the high  $\text{Fe}_2\text{O}_3$  content, there is no efficient embedding of the particles within the pores of the silica matrix. Thus, the oxide phase  $\gamma\text{-Fe}_2\text{O}_3$  is much more exposed to the transformation to  $\alpha\text{-Fe}_2\text{O}_3$ . It is notable that in this case, the  $\alpha\text{-Fe}_2\text{O}_3$  is stabilized at high temperatures without transformation to other phases. For all compositions  $B_x$ ,  $E_x$  where  $\gamma\text{-Fe}_2\text{O}_3$  was well crystallized independent on the annealing temperature, we have calculated the size of the nanoparticles from the (311) and (440) diffraction peaks using the Scherrer formula,<sup>21</sup> and we have obtained the mean diameter of  $\sim 5$  nm.

Figure 10 presents the XRD patterns of the gels  $B_{20}$ ,  $B_{30}$ ,  $B_{50}$  and  $B_{70}$  at 1073 K. **The patterns of the samples with 20, 30 and 50 wt%  $\text{Fe}_2\text{O}_3/\text{SiO}_2$  reveal the well crystallized  $\gamma\text{-Fe}_2\text{O}_3$  phase, but for the 50 wt%  $\text{Fe}_2\text{O}_3/\text{SiO}_2$ ,**

**the diffraction peak at 17° indicates the appearance of a weakly crystallized  $\epsilon\text{-Fe}_2\text{O}_3$  phase.** The XRD pattern of the sample with 70 wt%  $\text{Fe}_2\text{O}_3/\text{SiO}_2$  presents the  $\alpha\text{-Fe}_2\text{O}_3$  as the single crystalline phase.

Figures 11 and 12 present the TEM images of the samples  $B_{30}$  and  $B_{50}$  annealed at 1073 K. In the sample  $B_{30}$  (30%  $\gamma\text{-Fe}_2\text{O}_3/\text{SiO}_2$ ), the  $\gamma\text{-Fe}_2\text{O}_3$  nanoparticles are spherical and homogeneously dispersed within the silica matrix, with diameter of  $\sim 10$  nm. For the sample  $B_{50}$  (50%  $\gamma\text{-Fe}_2\text{O}_3/\text{SiO}_2$ ), the  $\gamma\text{-Fe}_2\text{O}_3$  nanoparticles are more agglomerated, of spherical shape with diameters in the range 10–12 nm.

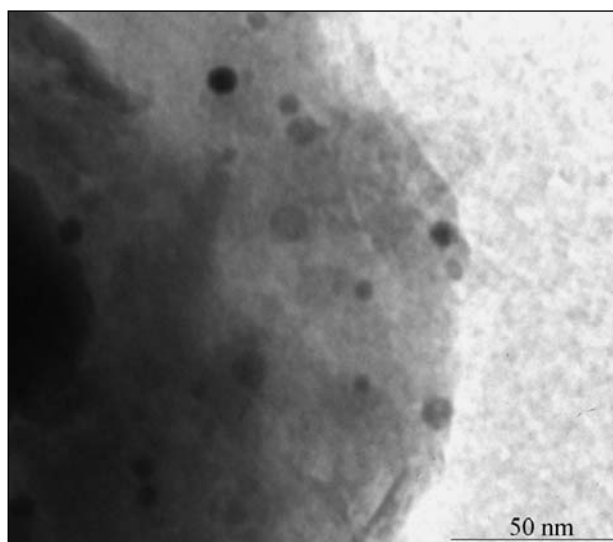


Figure 11. TEM image of the sample  $B_{30}$  annealed at 1073 K

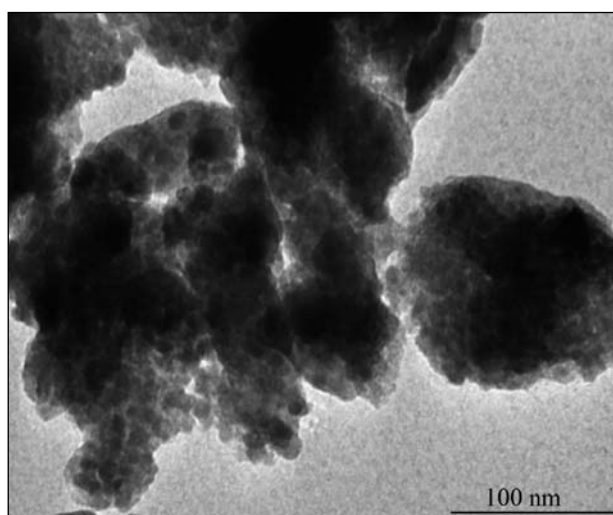


Figure 12. TEM image of the sample  $B_{50}$  annealed at 1073 K

In Figure 13 we present the magnetization curves of samples  $B_{50}$  annealed at 773 and 973 K, measured at room temperature. For both samples the coercive field is  $H_c = 0$

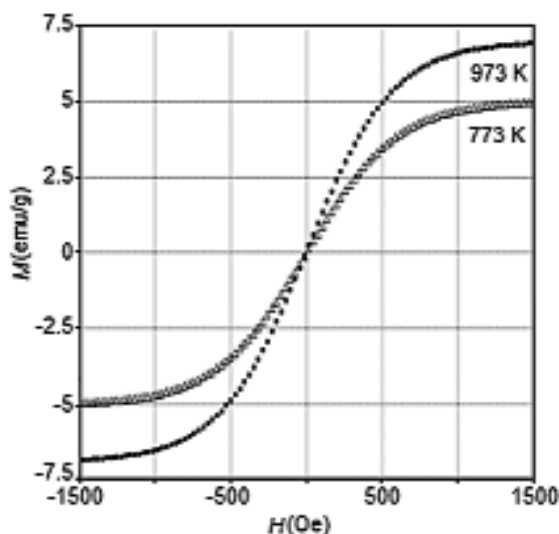


Figure 13. Magnetization curves of the sample B<sub>50</sub>

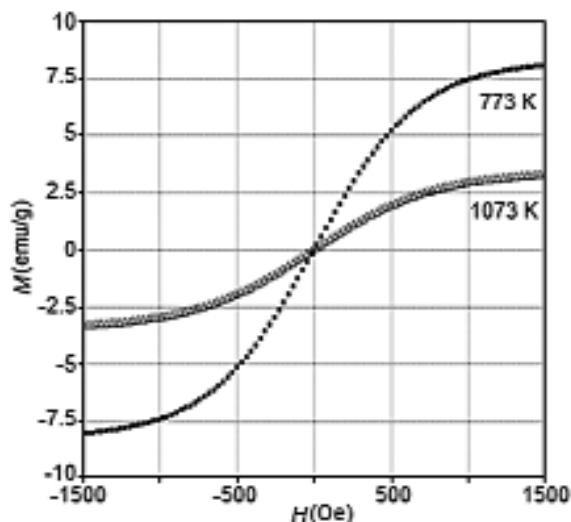


Figure 14. Magnetization curves of the sample B<sub>70</sub>

leading to the superparamagnetic behavior. The value of the saturation magnetization (emu/g) increases with temperature due to the crystallinity of  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>.

A higher  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub> content in the sample B<sub>70</sub> (Figure 14) leads to the saturation magnetization increase compared to the sample B<sub>50</sub> (Figure 13), at the same annealing temperature (773 K),  $H_c = 0$ , and the behavior is superparamagnetic.

It is notable that the sample B<sub>70</sub> annealed at 1073 K exhibits a magnetic behavior although the XRD pattern (Figure 9) reveals the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub> (antiferromagnetic) as the only phase. This can be explained by the fact that  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> is present in the matrix in a very low amount and it is of nanometric dimensions. Magnetic measurements agree with the XRD results for the samples B<sub>50</sub> and B<sub>70</sub>.

## 4. Conclusions

We report a new method to obtain the single phase  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub> in the temperature range 573–1073 K. The nature of the carboxylate precursors (glyoxylate, succinate) and the reducing environment created by their decomposition in the pores of the gel, influence the formation and crystallization of the  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles.

For ratios  $\leq 50$  wt% Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub>,  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles are formed and are stable up to 1073 K. In the cases when the samples are annealed at higher temperatures (1173 K), we observe the transformation of the  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> to the pure oxide phase  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub>. For higher ratios (70 wt%) the crystalline  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> phase is stable up to 773 K. At higher temperatures, the  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> transforms to the well crystallized stable  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub>.

The silica matrix remains amorphous even at high temperatures (1273 K).

The method allows obtaining the ferrimagnetic  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub> phase at high temperatures as well as a homogenous distribution of the nanoparticles even at high Fe<sub>2</sub>O<sub>3</sub>/SiO<sub>2</sub> ratios.

## 5. Acknowledgement

This work was supported by the National Project no. 71–026 NANOPART, Romanian Ministry of Education and Research.

## 6. References

1. S. Onodera, H. Kondo, T. Kawana, *MRS Bull.* **1996**, *21*, 35–40.
2. R. F. Ziolo, E. P. Giannelis, B. A. Weinstein, M. P. O'Horo, B. N. Ganguly, V. Mehrotra, M. W. Russel, D. R. Huffman, *Science* **1992**, *257*, 219–222.
3. R. D. McMichael, R. D. Shull, L. J. Schwartzendruber, L. H. Bennett, R. E. Watson, *J. Magn. Magn. Mater.* **1992**, *111*, 29–33.
4. I. J. Bruce, J. Taylor, M. Todd, M. J. Davies, E. Borioni, C. Sangregorio, T. Sen, *J. Magn. Magn. Mater.* **2004**, *284*, 145–160.
5. C. R. F. Lund, J. A. Dumesic, *J. Phys. Chem.* **1982**, *86*, 130–135.
6. R. Y. Hong, H. P. Fu, G. Q. Di, Y. Zheng, D. G. Wei, *Mat. Chem. Phys.* **2008**, *108*, 132–141.
7. N. Hüsing, U. Schubert, *Angew. Chem. Int. Ed.* **1998**, *37*, 22–45.
8. J. L. Dormann, *Mater. Sci. Eng. A* **1993**, *168*, 217–224.
9. L. I. Casas, A. Roig, E. Molins, J. M. Greneche, J. Asenjo, J. Tejada, *Appl. Phys. A* **2002**, *74*, 591–597.
10. J. M. Xue, Z. H. Zhou, J. Wang, *Mater. Chem. Phys.* **2002**, *75*, 81–85.
11. M. Birzescu, M. Cristea, M. Stefanescu, Gh. Constantin, *Ro. Pat.* 102501, **1990**.

12. C. Caizer, M. Stefanescu, *J. Phys. D: Appl. Phys.* **2002**, *35*, 3035–3040.
13. M. Stefanescu, O. Stefanescu, M. Stoia, C. Lazau, *J. Therm. Anal. Calorim.* **2007**, *88*, 27–32.
14. O. Stefanescu, C. Davidescu, M. Stefanescu, M. Stoia, *J. Therm. Anal. Calorim.* **2009**, *97* (1), 203–208.
15. M. Stefanescu, C. Caizer, M. Stoia, O. Stefanescu, *Acta Mater.* **2006**, *54*, 1249–256.
16. K. Nakamoto, Y. Morimoto, A. E. Martell, *J. Am. Chem. Soc.* **1961**, *83*, 4528–4532.
17. M. Stefanescu, M. Stoia, O. Stefanescu, *J. Sol-Gel Sci. Technol.* **2007**, *41*, 71–78.
18. D. Knetsch, W. L. Groeneveld, *Inorg. Chim. Acta* **1973**, *7*, 81–87.
19. R. Schrader, G. Buttner, *Z. Anorg. Allgem. Chem.* **1963**, *320*, 220–234.
20. P. Brázda, D. Niždanský, J.-L. Rehspringer, J. P. Vejpravova, *J. Sol-Gel Sci. Technol.* **2009**, *51*, 78–83.
21. R. Jenkins, R. L. Snyder, *Introduction to X-ray Powder Diffractometry*, John Wiley & Sons Inc., New York, **1996**, pp. 89–91.

## Povzetek

V delu opisujemo sintezo in temperaturno obstojnost nanodelcev  $\gamma\text{-Fe}_2\text{O}_3$  v gelih  $\text{SiO}_2$ . Slednja je odvisna od vrste Fe(III) karboksilatnega prekursorja, temperature segrevanja in koncentracije  $\text{Fe}^{3+}$ . Priprava prekursorjev v porah gela temelji na redoks reakciji med  $\text{Fe}(\text{NO}_3)_3$  in dioloma etilenglikolom (EG) in 1,4-butandiolom (1,4 BG). Pripravili smo gele z različnimi razmerji  $\text{Fe}_2\text{O}_3/\text{SiO}_2$  (20, 30, 50, 70 utežnih %  $\text{Fe}_2\text{O}_3$ ) in jih segrevali pri temperaturah med 573 in 1273 K.

Na nastanek in stabilnost enofaznega  $\gamma\text{-Fe}_2\text{O}_3$  v porah gela močno vpliva redukcijska atmosfera, ki se vzpostavi med termičnim razpadom Fe(III) karboksilatov. Rentgenska fazna analiza je potrdila izrazitejšo kristalizacijo  $\gamma\text{-Fe}_2\text{O}_3$ , ko je bil uporabljen 1,4 BG. Magnetne meritve so potrdile kristalinično fazo  $\gamma\text{-Fe}_2\text{O}_3$  v matrici silicijevega oksida.